

Hybrid Directed Energy Deposition of Geometrically-Complex Pressure Vessels for Advanced HIP Canning and Digitally-Driven Powder Metallurgy

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Abstract

Directed Energy Deposition (DED) is one of the highest production rate additive manufacturing processes, yet it often faces challenges with dimensional accuracy and surface finish compared to powder bed fusion. Fabricating structures with internal cavities using DED is typically constrained unless a five axis motion system is employed, allowing for the maintenance of a normal to gravity orientation during deposition, which is essential for creating overhanging features and completing enclosed cavities. This capability is particularly valuable for applications such as pressure vessels and geometrically complex hot isostatic pressing (HIP) containers. The present work explores the tool path strategy required to construct a toroidal cylinder with an internal cavity. Leak testing and cross-sectional analysis such as optical microscopy and the Archimedes density test, confirm that the structure is sufficiently pressure-tight and exhibits minimal porosity, making it suitable for use as complex HIP can, thus paving the way for the next generation of multi-material, digitally-driven powder metallurgy.

Keywords: Hybrid Manufacturing, Directed Energy Deposition (DED), HIP Canning, Powder Metallurgy, Additive Manufacturing,

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1. Introduction

Manufacturing is evolving to leverage simultaneously the benefits of additive manufacturing (AM) - such as geometric freedom, part customization, and low waste—with the dimensional accuracy and surface finish of machining [1–4]. In hybrid manufacturing, additive and subtractive processes are integrated within a single build chamber, allowing for the creation of features that would be difficult or impossible to achieve with either method alone [5–7]. Examples include internal cavities with machined surfaces [8] and internally embedded components [9, 10].

Directed Energy Deposition (DED), the AM process employed in this study, has been extensively explored since the mid-1990s [11–15] for building metal, ceramic, and polymer structures [6, 16–23]. DED uses powder or wire feedstock melted by a laser, electric arc, or electron beam [24]. In powder-based systems, powder is delivered to the deposition point via a focused inert gas flow, coinciding with a laser source that melts the powder onto the structure [18]. The result is a strong metallurgical bond between layers or to the substrate, offering full spatial control for on-demand alloying, surface coatings, and component repair [21, 25–28].

Hot Isostatic Pressing (HIP) is a commonly used post-processing technique, especially in the qualification of aerospace structures made via AM. HIP enhances the quality of printed structures, particularly those with complex geometries formed by electron beam [29–37] or laser [38–46] Powder Bed Fusion (PBF), as well as DED [47–51] and Material Extrusion [52]. By applying high temperature and pressure simultaneously, HIP densifies structures and homogenizes their microstructure [53]. HIP is effective in closing large pores (up to 3 mm in diameter) in Ti6Al4V and steel structures made by both traditional and additive methods [54–59]. The increased density from HIP improves fatigue performance [60], making it a standard final process for additively manufactured aerospace components [29, 61]. The concept of building HIP cans using AM has been investigated, particularly through PBF. In these studies, shell geometries were fabricated with internal cavities containing trapped, unmelted powder [62, 63], and included the evaluation of fatigue properties [64]. However, these works were limited to a single-material powder, matching the shell material. PBF is currently specialized for small to medium-sized components due to production speed constraints, but HIP offers more significant advantages for large components. In this context, hybrid DED excels, as it can produce larger components with material versatility, though it may require additional post-processing to achieve the desired surface finish and detail. Finally, as laser PBF is not typically performed in a hard vacuum, gasses were included with the trapped powder, which could lead to thermally induced porosity after subsequent high temperature exposure [65, 66]. Further considerations for AM-enabled HIP canning include the requirement for can materials to have good ductility to distort with the consolidation of internal powder but also be sufficiently strong to maintain the pressure.

The present work investigates the capability of hybrid AM to build a metallic pressurized vessel to enable applications such as rocket fuel tanks, but more specifically for enhancing powder metallurgy by fabricating geometrically complex HIP containers. The digital-driven paradigm of AM can now enable new geometries from CAD directly for use as HIP cans - not be easily fabricated with traditional approaches. Furthermore, AM HIP cans can include features to facilitate the introduction of powders with pre-attached fill

tubes, simulation-driven geometry compensation for HIP deformation, varying thickness walls to optimize the final geometry, and multiple chambers to introduce different metals geometrically. These structures, if capable of holding high pressures without leakage, can be: (a) filled with metal powder; (b) evacuated without disturbing the internal powder to reduce thermally induced porosity, and then finally; (c) welded hermetically closed in preparation of HIP for comprehensive consolidation. Consequently, AM can enable a new generation of powder metallurgy in which HIP canning can provide multi-material unique geometries.

2. Methodology

2.1. Materials

AISI Type 316L Stainless Steel (SS316L) was selected for this work due to its relevance in several industrial applications [67] and well-documented properties in prior AM research [22, 68]. SS316L powder was deposited onto a cylindrical (155 x 50 mm) SS316L substrate. The gas-atomized powder (Oerlikon Metco (US) Inc.) had a particle size distribution with percentiles of $D_{10} = 54.02 \mu\text{m}$, $D_{50} = 73.35 \mu\text{m}$ and $D_{90} = 114.50 \mu\text{m}$. Helium was selected as the carrier gas for the powder due to its availability in the machine setup.

2.2. Manufacturing Procedure

The component was manufactured on an Okuma MU8000V-L Laser EX hybrid manufacturing system (Figure 1). The DED fabrication process was carried out using parameters established in previous studies, as detailed in Table 1. The internal cylinder was fabricated using an additive turning toolpath, developed by OPEN MIND Technologies AG as a component of its hyperMILL CAM software. The additive turning toolpath allows axisymmetric components with fluctuating wall thickness to be manufactured by employing a combined raster-spiral method.

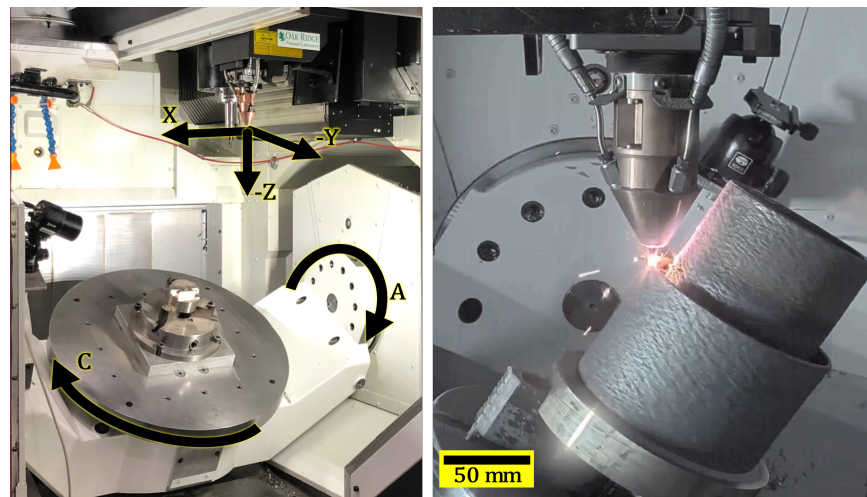


Fig. 1 Okuma MU8000V-L LASER EX five-axis hybrid blown-powder directed energy deposition system (left) and toroidal structure with internal cavity in fabrication (right).

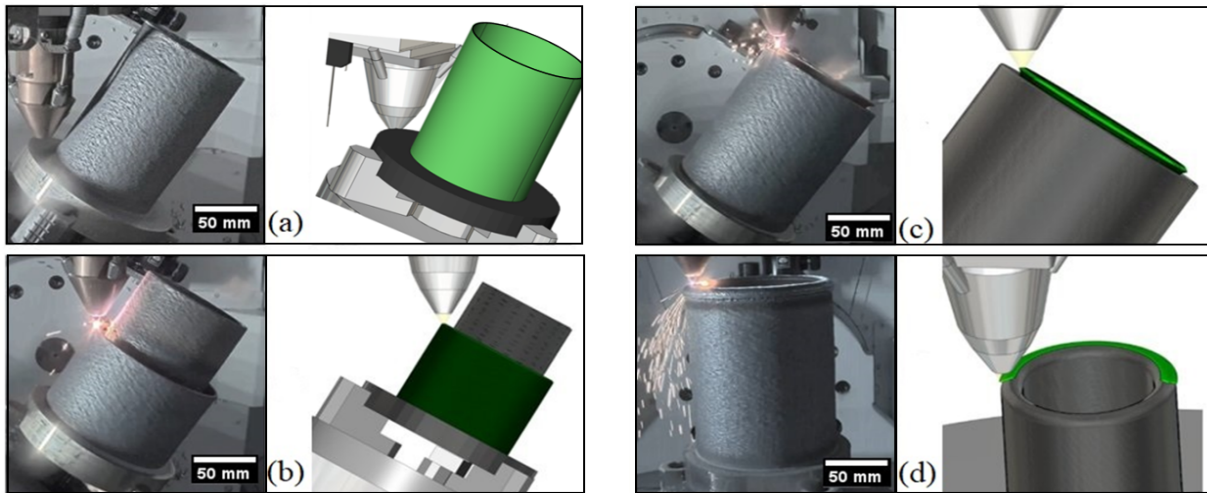
The component was designed to have a height of 165mm, a wall thickness of 4mm, and an inner diameter of 95mm. For the additive turning parameters, the pitch-X and pitch-Z were 2.4mm and 0.66mm [69]. For the outer cylinder, the deposition angle was

Table 1 DED Process Parameters

Parameter	LP-DED
Laser spot diameter (mm)	3.5
Laser power (W)	2,000
Powder feed rate (g/min)	8.6
Traverse feed rate (mm/min)	500*
Layer height (mm)	0.35
Argon shield flow rate (l/min)	10
Helium carrier flow rate (l/min)	5

*Unless noted otherwise

tilted to 30° to avoid obstruction (Figure 2b). Since the outer cylinder was printed at an oblique angle, the traverse speed was adjusted to maintain the proper standoff or distance between the deposition nozzle and deposition surface. An initial traverse feed rate of 900 mm/min was used but was manually adjusted between 500 mm/min and 900mm/min until stabilizing on a final traverse feed rate of 700 mm/min. These adjustments are common in many DED systems and can be detrimental to both geometrical accuracy and metallurgical properties of the final component. However, for HIP cans, the authors expect these changes to be irrelevant if the HIP can hold a vacuum. As the HIP can be removed upon final machining, the minor variance in the processing conditions is expected to be inconsequential.

**Fig. 2** Steps in the fabrication: a) first interior cylinder; b) second exterior cylinder; c) first of two half arcs; and d) final deposition on the top seam.

The final steps include maintaining a normal with gravity and creating a half arc from the interior cylinder top to span halfway to the exterior cylinder (Figure 2c and Figure 3). A second arc is performed from the external cylinder back to the interior at a rate of 650 mm/min to complete the arch to serve as an unsupported roof of the now cavity. A final additional layer was deposited using a weaving strategy between the two half arcs to close any areas that did not fully fuse and to reinforce the arc structure with a seam at a feed rate of 500 mm/min (Figure 2d). This weave strategy used a pitch of 10 mm and a pitch of 2 mm. The resulting structure, including the build plate, now contains a cylindrical internal chamber which was hypothesized to be capable of handling a vacuum. Additional features such as feed tubes or additional machining could be included for smooth surfaces

or prescribed wall thickness as is required. Future work will investigate these and other advantages for HIP for advanced geometrically complex powder metallurgy.

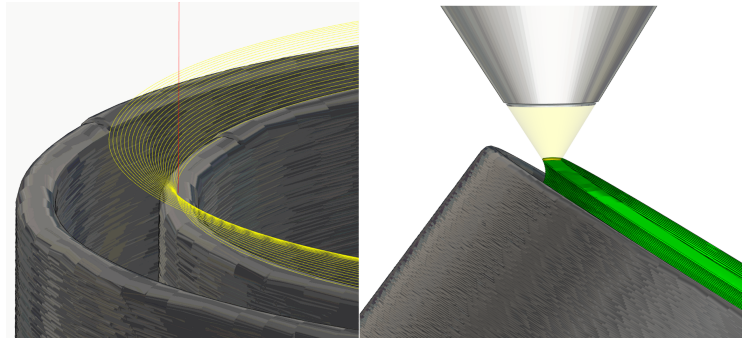


Fig. 3 Tool path for the first arc of the cavity roof printed with constant normal to gravity and schematic of deposition. Path (left) and deposition simulation (right). Yellow lines correspond to G1 (deposition) movement, while the red line corresponds to G0 (reposition) movement.

2.3. Leakage Testing

The vacuum and pressure integrity of the additively manufactured HIP container was verified by methodically spraying helium gas from a nozzle over the container and sealing flange assembly shown in Figure 4. The sealing flange arrangement was made by tapping a 2mm hole into the build plate and then mounting a face flange and O-ring to make the effect of a vacuum seal. The seal was then mated to a Swagelok tube and bellows connected to a Varian MD30 helium leak detector.



Fig. 4 Vacuum testing setup with DED manufactured cylinder with connected baseplate on top.

2.4. Density and Porosity Characterization

After pressure testing, the container was cross sectioned in half along the XZ plane using a bandsaw. Subsequently, a 30 mm thick slice was removed, and then three samples

were obtained by making cuts at 40 mm, 56.5 mm, and 78.5 mm from the slice's top surface, producing a double-walled 40 mm tall arch (sample 1), a 16.5 mm tall external wall (sample 2), and a 22 mm tall external wall (sample 3).

The density and porosity of the samples was evaluated via Archimedes' suspension method [70] and micrographic analysis. Archimedes' method was performed using ethanol at 23 °C with an Ohaus Explorer balance and densities were reported as a percentage of the 8.0 g/cm³ theoretical density (TD) for SS316L [71, 72]. Samples 1 and 3 were then mounted in epoxy, ground using 500, 800, 1000, and 2000 grit silicon carbide discs in 1 min intervals with water as the lubricant, and then polished using 6, 3, and 1 μm diamond solution with DP-Lubricant Purple in 8 min intervals.

The samples were optically imaged using a Leica DM4000M microscope at a resolution of 1 μm/pixel, employing a brightfield contrasting method with a 5X magnification. The Leica Application Suite X software automatically captured and stitched images of the entire mounted cross-sections into mosaics, enabling the analysis of statistically significant sample sizes. These images were then processed and analyzed using ImageJ®[73]. For porosity characterization, the images were binarized by thresholding to measure pore area, size, and total porosity area fraction (black-to-total pixel ratio). Special focus was placed on the quality of the cavity-roof arch, which was difficult to fabricate due to the oblique deposition angle used to align the DED process with gravity and avoid internal supports.

3. Results and Discussion

The most challenging aspect of this structure was after the completion of the two independent cylinders. Two half arcs were built to span between the cylinders and to provide a sealed cavity. The arcs were created using the tool path shown in Fig. 3, resulting in a non-closed surface remaining after the two operations. A final bead was then deposited to complete the full arch and to provide a sealed and pressure-capable cavity as shown in Fig. 5.



Fig. 5 Fabrication results: (left) after completing inner half arch; (middle) after completing both arches but with gaps remaining; (right) after final bead for complete seal.

3.1. Leakage Testing

Measurements on the sealed additively manufactured container showed a helium leak rate of $\sim 1.0 \times 10^{-9}$ Nml/min or less, indicating sufficient integrity for subsequent HIP trials. By spraying helium around the structure during the cavity vacuum pull, any

leaks can be detected, and their locations identified. Future studies will incorporate machining to create precise, thin walls to enhance compliance and ductility for use as a HIP can. In this initial proof-of-concept demonstration, the structure successfully held a vacuum, suggesting that additively manufactured structures have potential for creating geometrically complex HIP cans. During the leak test, we also observed remnant powder from the deposition process inside the enclosed vessel. Despite efforts to remove the excess powder through the vacuum port, complete removal was challenging. This residual powder can damage vacuum test equipment, so future work will focus on developing strategies for powder removal and mitigation.

3.2. Density and Porosity Characterization

After completing the non-destructive leak testing, the part was cross-sectioned to assess the walls and top arch. The cross-section revealed no major defects or issues with the cavity's integrity (Fig. 6). The wall in this example was 3.5 mm thick with rough internal and external surfaces. One advantage of hybrid AM is the ability to machine both internal and external surfaces of cavities, allowing precise control over wall thickness and surface finish for HIP containers. Thinning the walls can enhance the structural ductility needed for effective HIP canning, enabling the container to compress fully and consolidate the internal powder. In contrast, excessively thick and stiff walls could hinder this consolidation during pressing.

AM alone, typically used for near-net shape fabrication, may not achieve the wall thickness precision required for HIP containers. However, by using in situ machining and HIP simulations, wall thickness can be varied to guide solidification shrinkage and ensure the final structure meets the intended geometry. This ability to vary wall thickness, along with creating complex geometric forms, is a key reason for using hybrid AM in the fabrication of HIP containers.

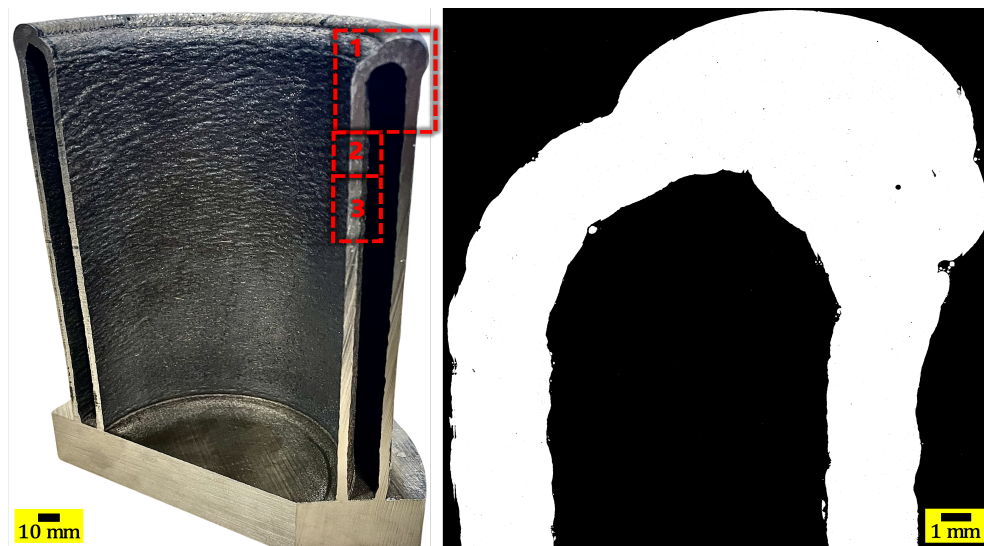


Fig. 6 Manufactured HIP-can : (left) after cross-sectioning, with numbers indicating the samples used for density and porosity characterization; (right) binarized optical micrograph of sample 1.

Based on the quantification of porosity area fraction, samples 1 and 3 were 99.90% and 99.96% dense, respectively. However, the densities derived via Archimedes' method for

samples 1-3 were 99.12%, 98.95%, and 99.12%, respectively (Table 2), which are slightly lower than the SS316L densities reported in the DED literature [74–76].

Table 2 Density and Percentage Values by Section

Section	Density (g/cm ³)	%
1	7.92	99.11
2	7.91	98.95
3	7.92	99.118

Sample 1 pores (1,802) measured $6.8 \pm 0.8 \mu\text{m}$ while sample 3’s (607) measured $6.4 \pm 0.4 \mu\text{m}$. Their corresponding pore size distributions (Fig. 7) showed exponential decay with approx. 95% of pores measuring $\leq 30 \mu\text{m}$. The remaining 5% of pores, ranging from 30.0-195.2 μm and 30.0-137.9 μm for samples 1 and 3, respectively, led to the lower than expected part density. Compared to sample 3, the higher frequency of pores $\leq 4 \mu\text{m}$ in sample 1 is attributed to its larger volume.

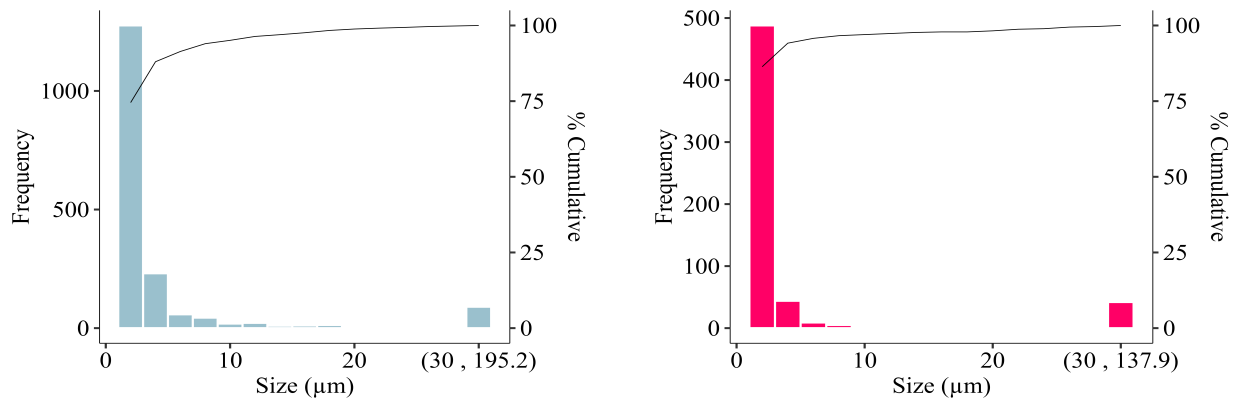


Fig. 7 Pore size distribution for sample 1 (left) and sample 3 (right). Sample numbers correspond to those shown in Fig. 6.

4. Conclusions

The present preliminary investigation demonstrates the potential of a hybrid additive manufacturing system in fabricating complex geometries with vacuum-capable internal cavities, specifically for HIP canning applications. The system’s five-axis capabilities successfully produced a structure with complex overhanging features, avoiding the need for support material and showcasing the feasibility of building sophisticated designs. Importantly, the HIP can was effectively designed using an additive turning toolpath, which facilitates the production of axisymmetric components with varying wall thicknesses through a combined raster-spiral technique. Although machining was not required in this proof of concept, the hybrid approach ensures that wall thickness can be meticulously controlled, with internal cavity surfaces accessible for machining during fabrication.

The fabrication process included challenges, particularly in sealing the internal cavity after building two independent cylinders. The application of a final bead after the construction of the half arcs successfully completed the seal, resulting in a pressure-capable cavity. Leak testing confirmed the integrity of the sealed structure, showing a helium

leak rate of $\sim 1.0 \times 10^{-9}$ Nml/min, which is sufficient for subsequent HIP trials. The implications of this work are significant for the future of HIP container manufacturing.

Hybrid fabrication enables:

1. Complex, digitally driven HIP can geometries supported by process simulation
2. Variable thickness walls to control and optimize warpage and shrinkage
3. Multiple chambers for multi-material powder metallurgy
4. Inclusion of HIP-can features such as feed tubes to easily weld and seal ports after introducing and evacuating powders and just prior to HIP densification of the final HIP can structure.

Despite the generally slower production rates of additive manufacturing compared to traditional sheet metal welding, the design freedom and digital precision offered by this approach introduce new possibilities for HIP can geometries that are otherwise unattainable. The hybrid process allows for the near-net shape fabrication of multi-chamber shells, with in situ machining to achieve precise dimensional accuracies. Additionally, HIP simulations can preemptively compensate for deformation and shrinkage, further enhancing the potential for complex, multi-material structures. The digitization of the HIP canning process marks a significant advancement in powder metallurgy, paving the way for the next generation of geometrically complex, high-temperature, multi-material structures. Ultimately, hybrid additive manufacturing not only enhances the precision and flexibility of HIP can fabrication but also optimizes consolidation during pressing through the ability to machine surfaces and control wall thickness. These advancements promise to improve yield, quality, and production rates for AM-enabled HIP canning, advancing the field of multi-material, digitally-driven powder metallurgy beyond current capabilities and potentially redefining economic benchmarks in industry.

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